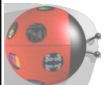


Introduction to Diffraction analysis

Luca Lutterotti

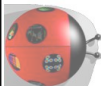
***Department of Materials Engineering
and Industrial Technologies***

University of Trento - Italy



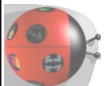
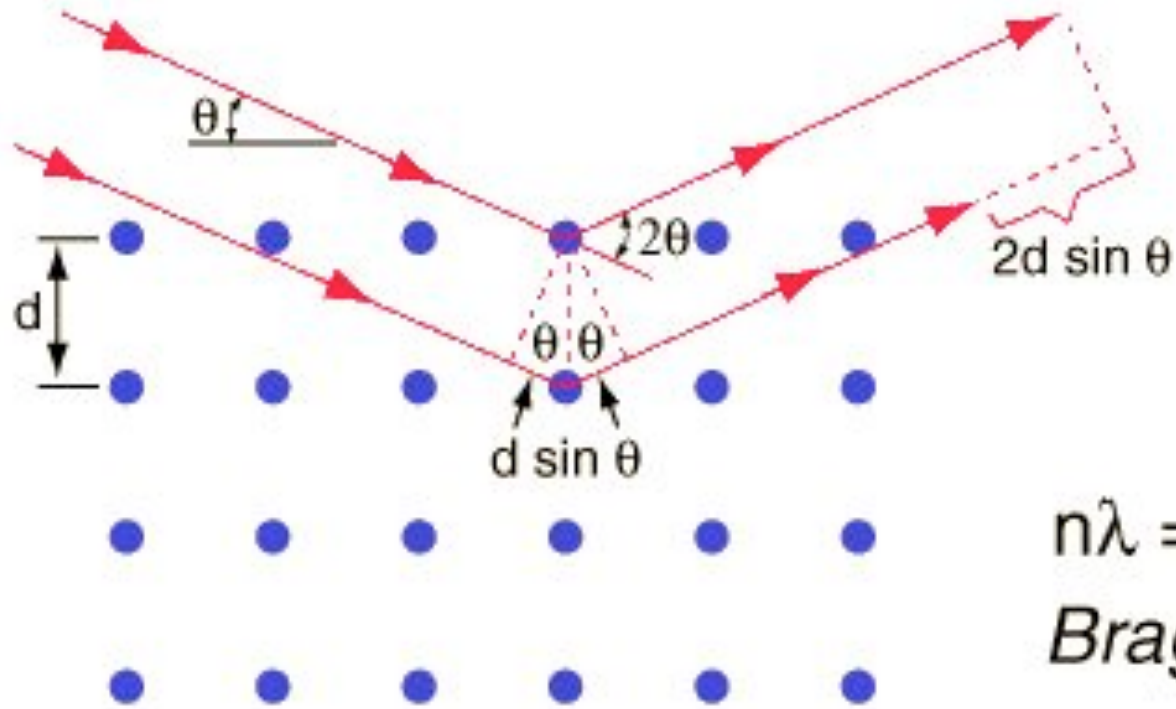
Outline: basic concepts

- *The Bragg law*
- *The intensity of the diffraction*
- *Powder diffraction and instrumentation*
 - *Bragg-Brentano*
 - *Texture goniometers*
 - *Residual stress measurements*
- *Diffraction analyses*



The Bragg law

- *Constructive interference and interplanar spacing:*

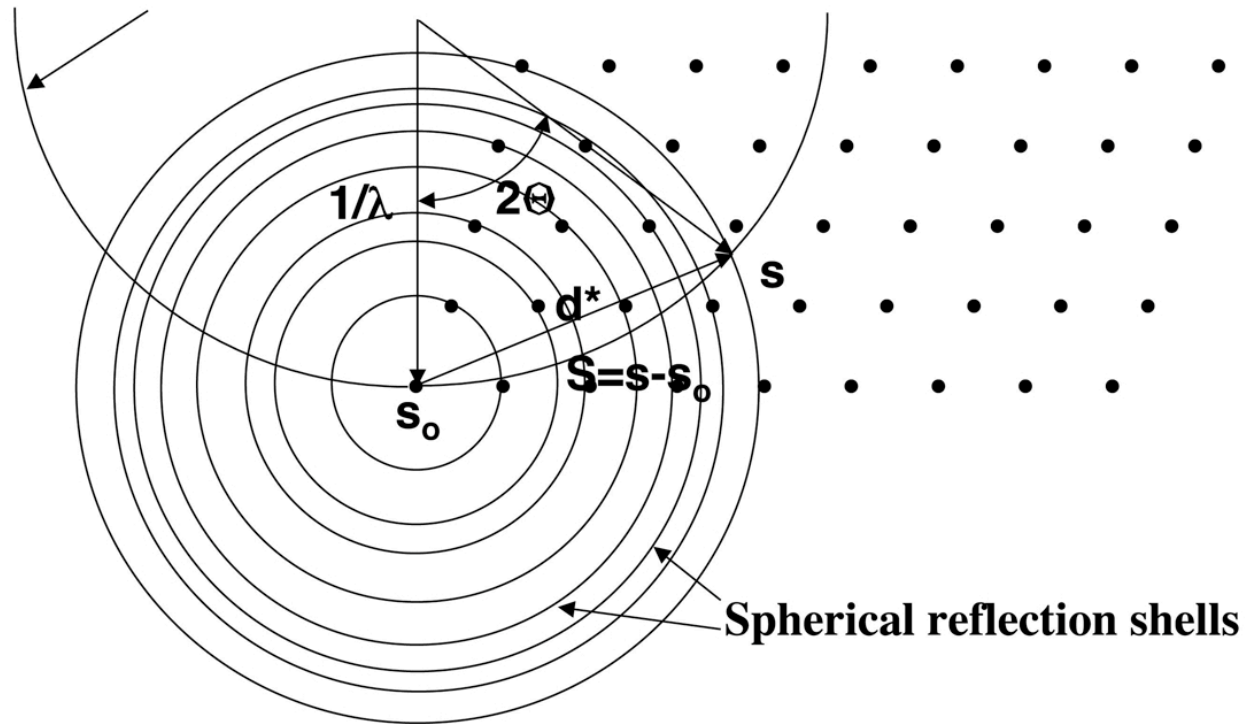


Reciprocal space

Again get Bragg's Law

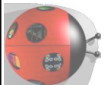
$$d^*/2 = \sin \Theta / \lambda$$

Ewald sphere



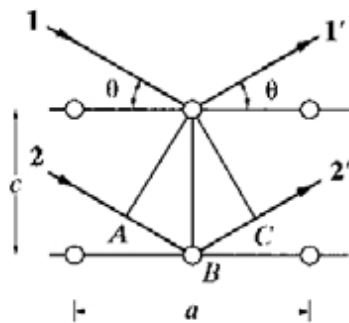
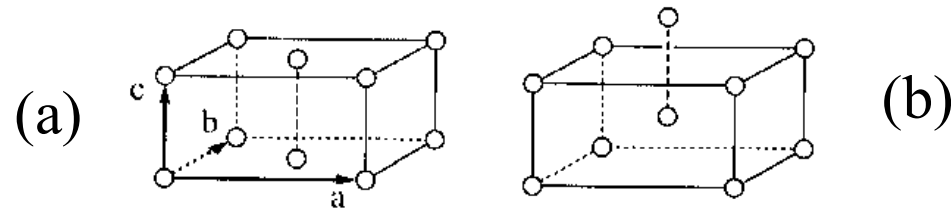
Spherical reflection shells

$$d_{hkl} = \frac{V_C}{\sqrt{s_{11}h^2 + s_{22}k^2 + s_{33}l^2 + 2s_{12}hk + 2s_{13}hl + 2s_{23}kl}}$$

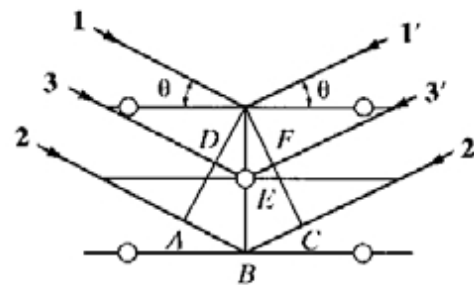


The Bragg law and the intensities

- Consider diffraction from the (001) plane

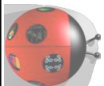


(a)



(b)

If the path length between rays 1 and 2 differs by λ , the path length between rays 1 and 3 will differ by $\lambda/2$ and destructive interference in (b) will lead to no diffracted intensity



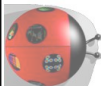
Diffraction intensities

- ***The intensity in a powder diffractometer***

$$I_i^{calc} = S_F \sum_{j=1}^{N_{phases}} \frac{f_j}{V_j^2} \sum_{k=1}^{N_{peaks}} L_k |F_{k,j}|^2 S_j(2\theta_i - 2\theta_{k,j}) P_{k,j} A_j + bkg_i$$

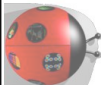
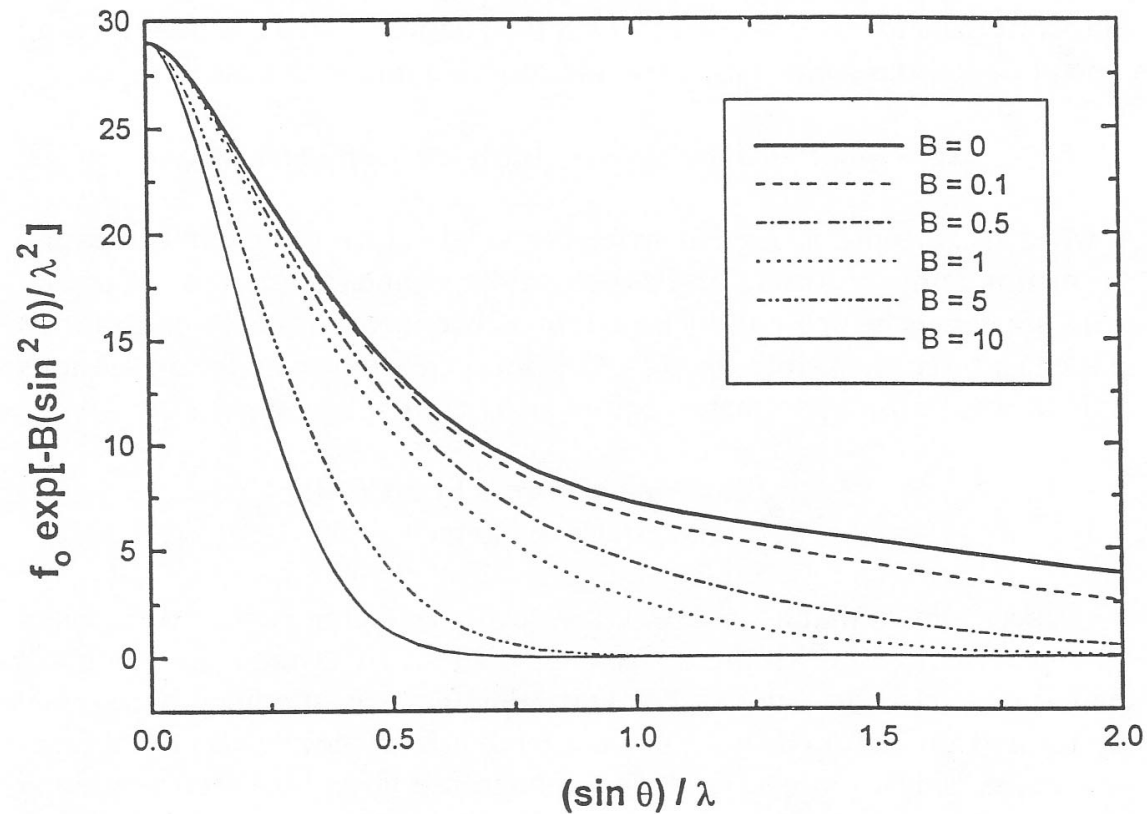
- ***The structure factor:***

$$|F_{k,j}|^2 = m_k \left| \sum_{n=1}^N f_n e^{-B_n \frac{\sin^2 \theta}{\lambda^2}} \left(e^{2\pi i (hx_n + ky_n + lz_n)} \right) \right|^2$$



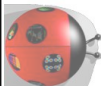
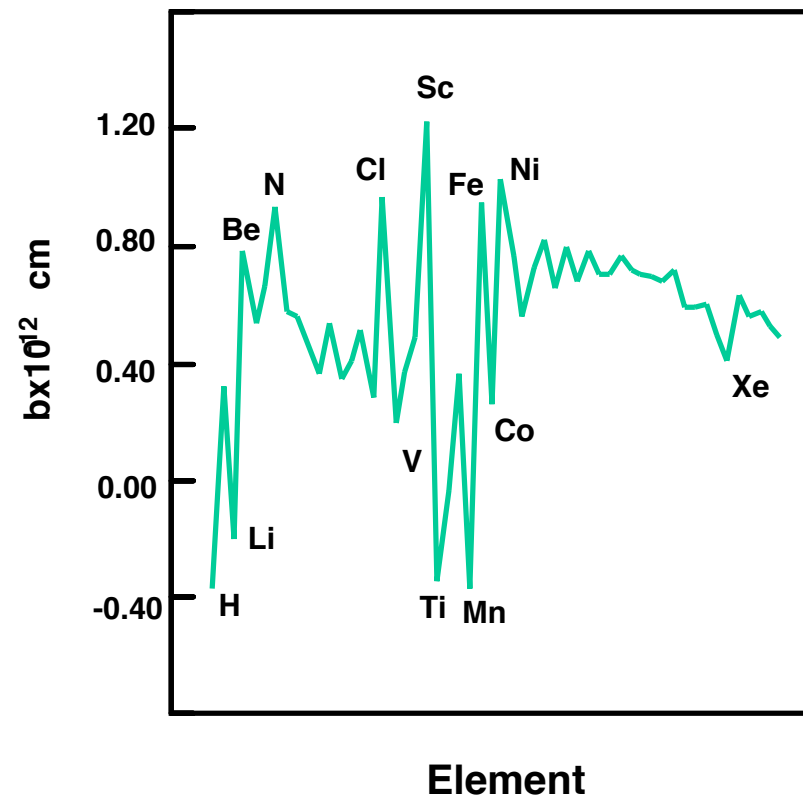
Atomic scattering factor and Debye-Waller

- The atomic scattering factor for X-ray decreases with the diffraction angle and is proportional to the number of electrons. For neutron is not correlated to the atomic number.



Neutron scattering factors

- *For light atoms neutron scattering has some advantages*
- *For atoms very close in the periodic table, neutron scattering may help distinguish them.*



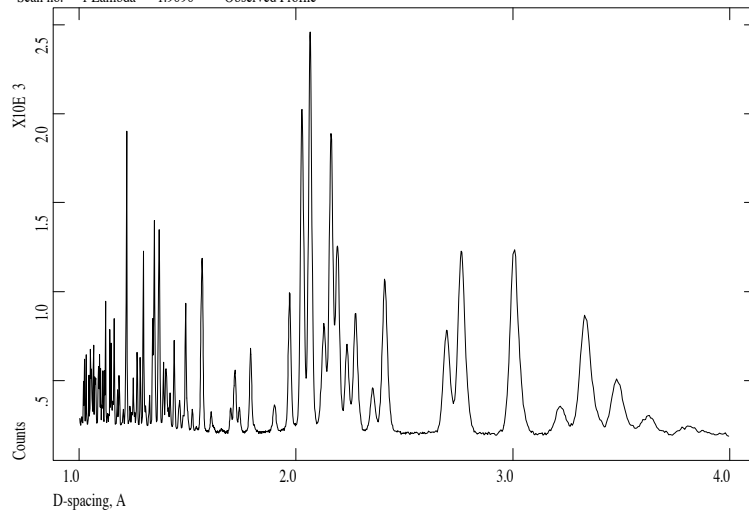
X-ray and neutron diffraction

X-ray Diffraction - CuK α Phillips PW1710

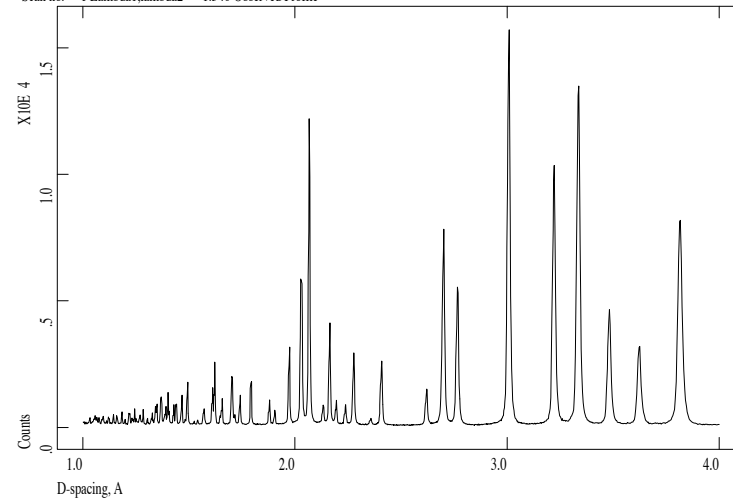
- Higher resolution
- Intensity falloff at small d spacings
- Better at resolving small lattice distortions



10.0 0.05 155.9 CPD RRRR PbSO4 1.909A neutron data 8.8
Scan no. = 1 Lambda = 1.9090 Observed Profile



10.000 0.025 159.00 CPD RRRR PbSO4 Cu K α X-ray data 22.9.
Scan no. = 1 Lambda1,lambda2 = 1.540 Observed Profile



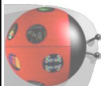
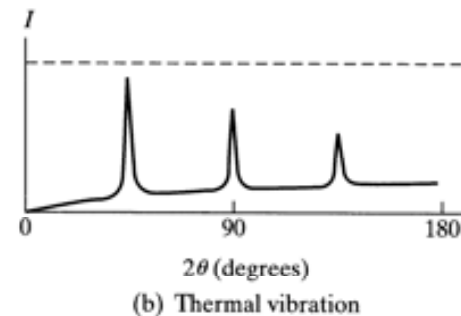
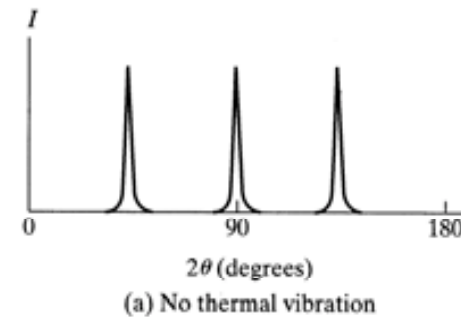
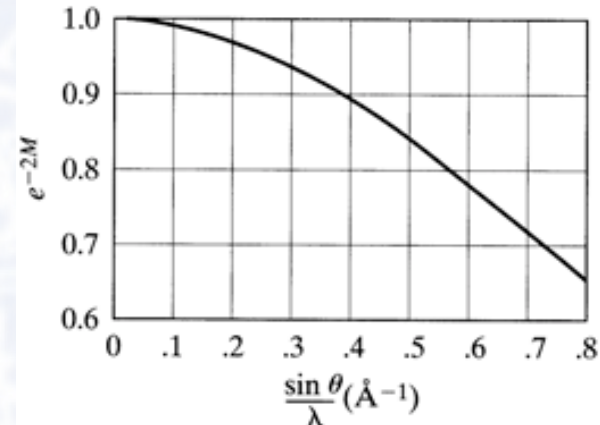
Neutron Diffraction - D1a, ILL $\lambda=1.909 \text{ \AA}$

- Lower resolution
- Much higher intensity at small d-spacings
- Better atomic positions/thermal parameters

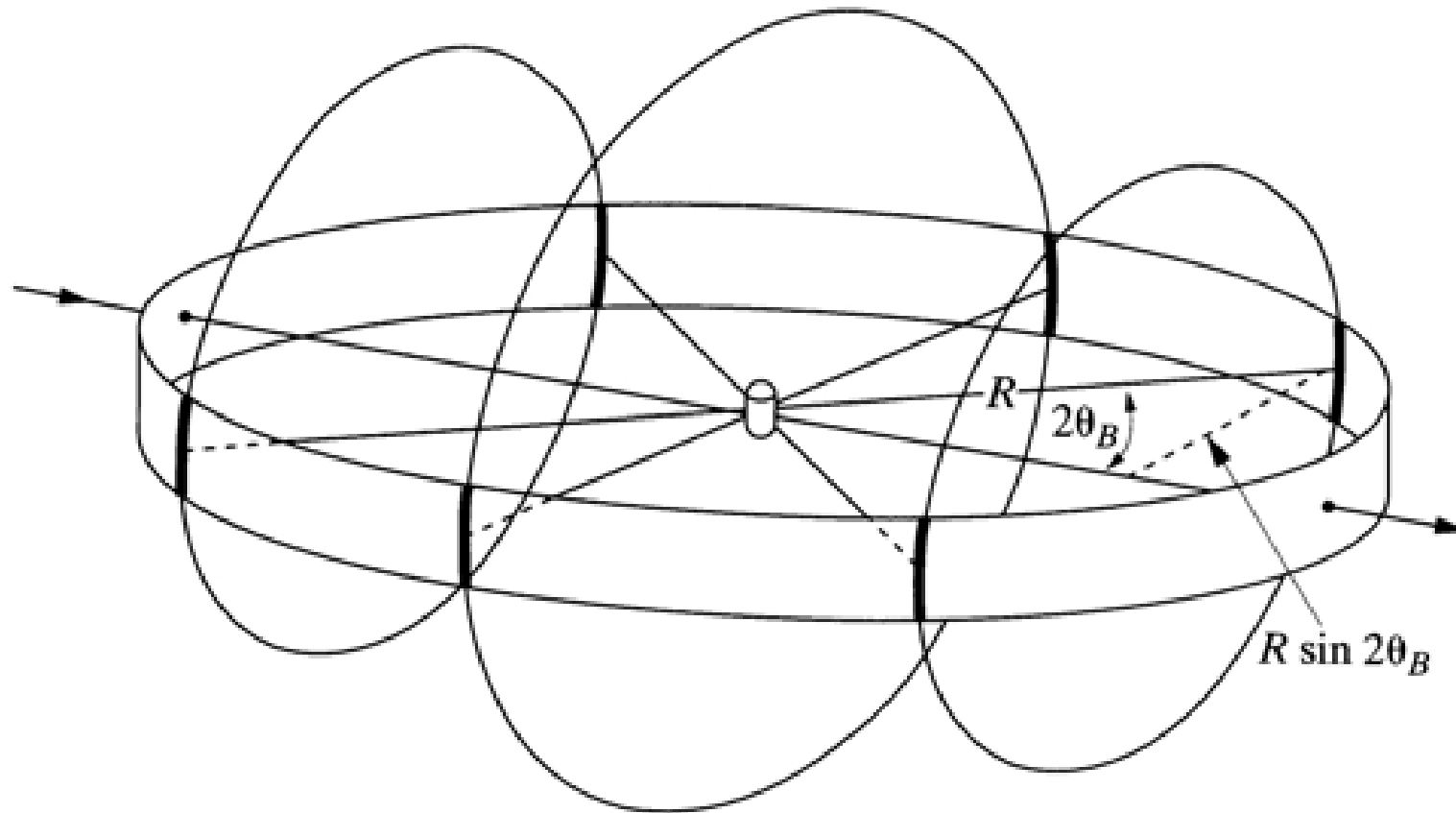


Thermal or Debye-Waller factor

- *It causes a decrease of the intensities at high angle*
- *It is proportional to the thermal vibrations*
- *Intensities decrease increasing the temperature*
- *From the Debye-Waller it is possible to estimate the Debye temperature*

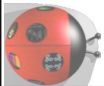
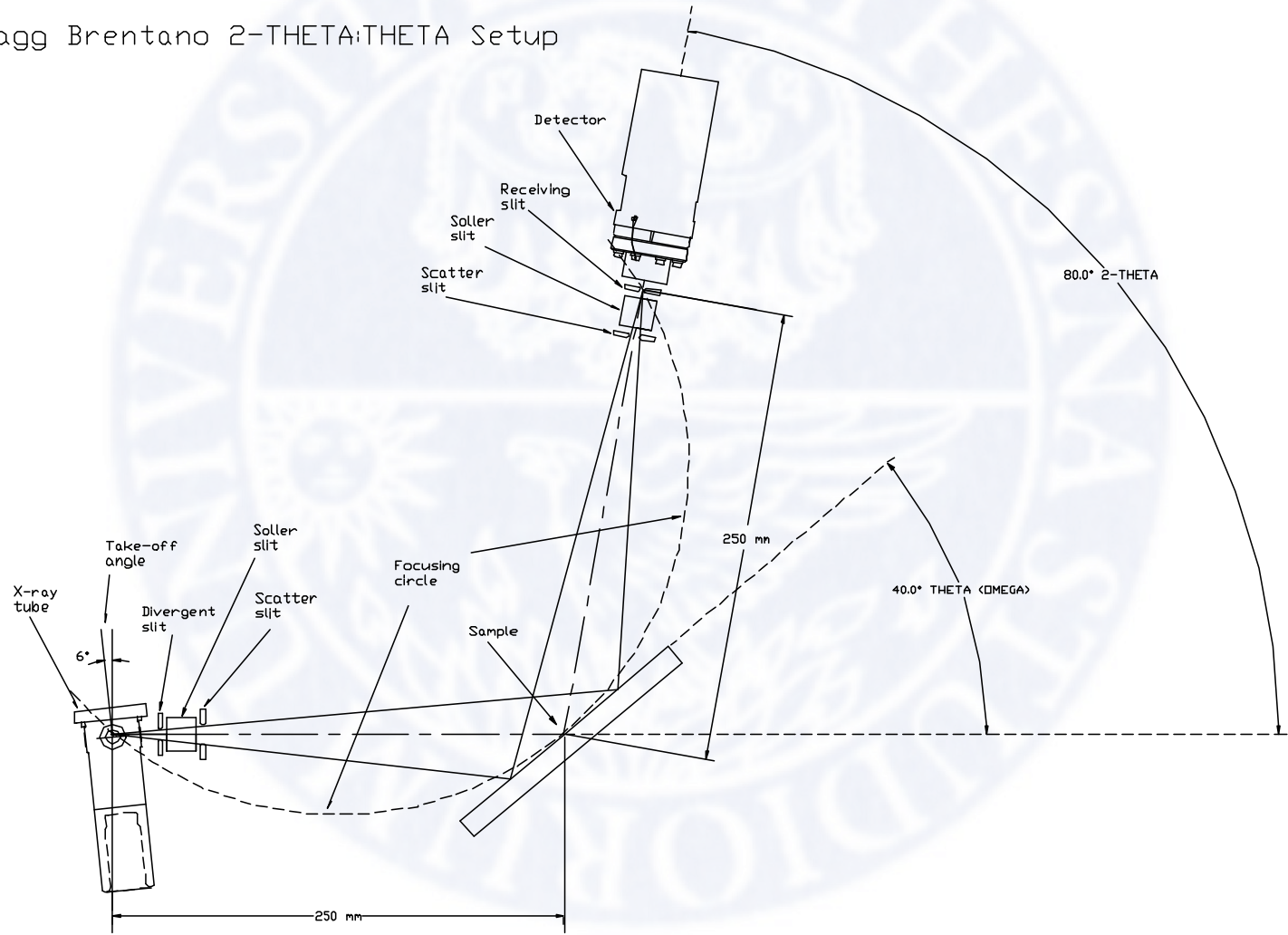


Powder diffraction and Debye-Scherrer camera

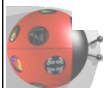
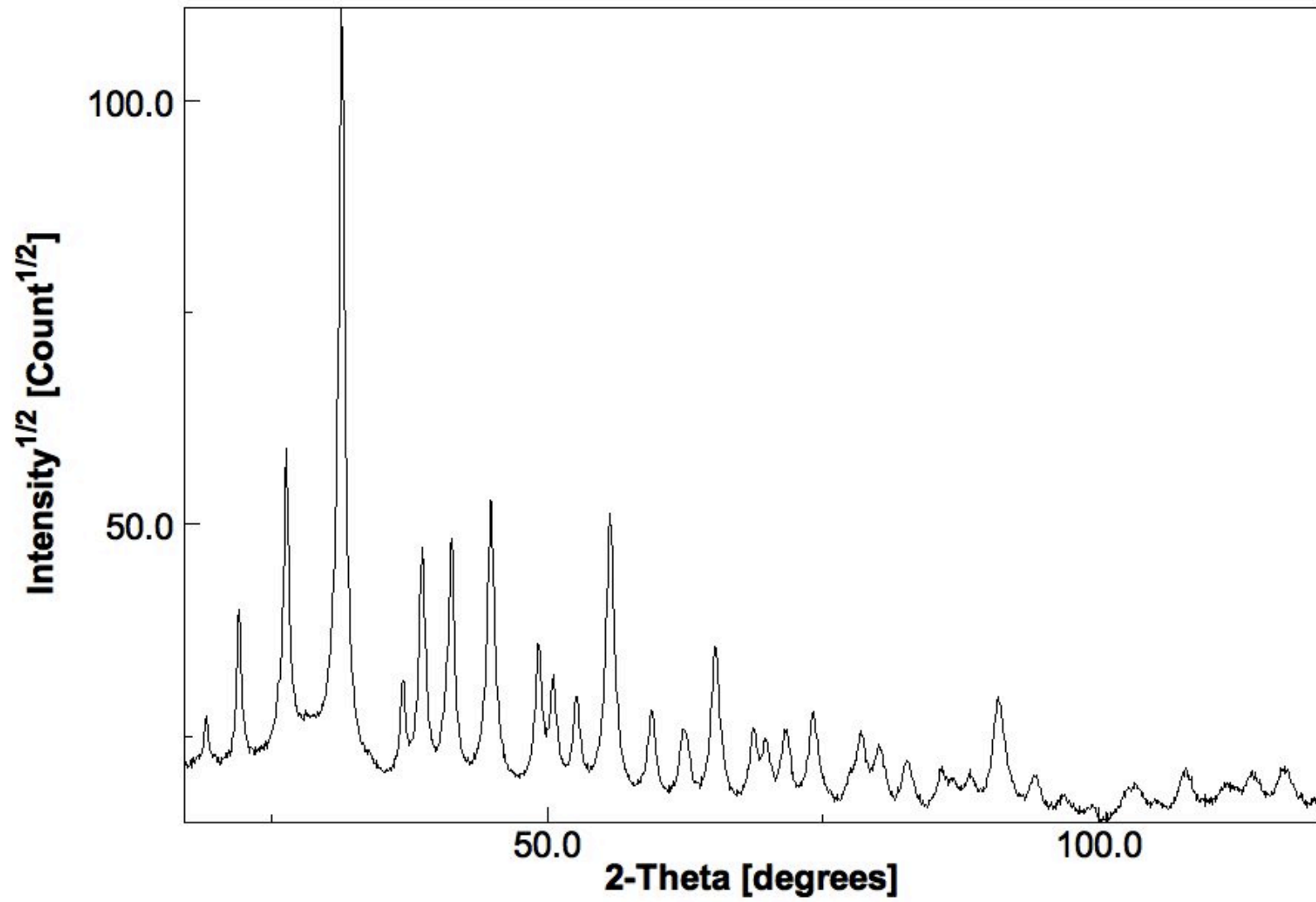


A modern diffractometer

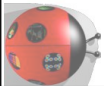
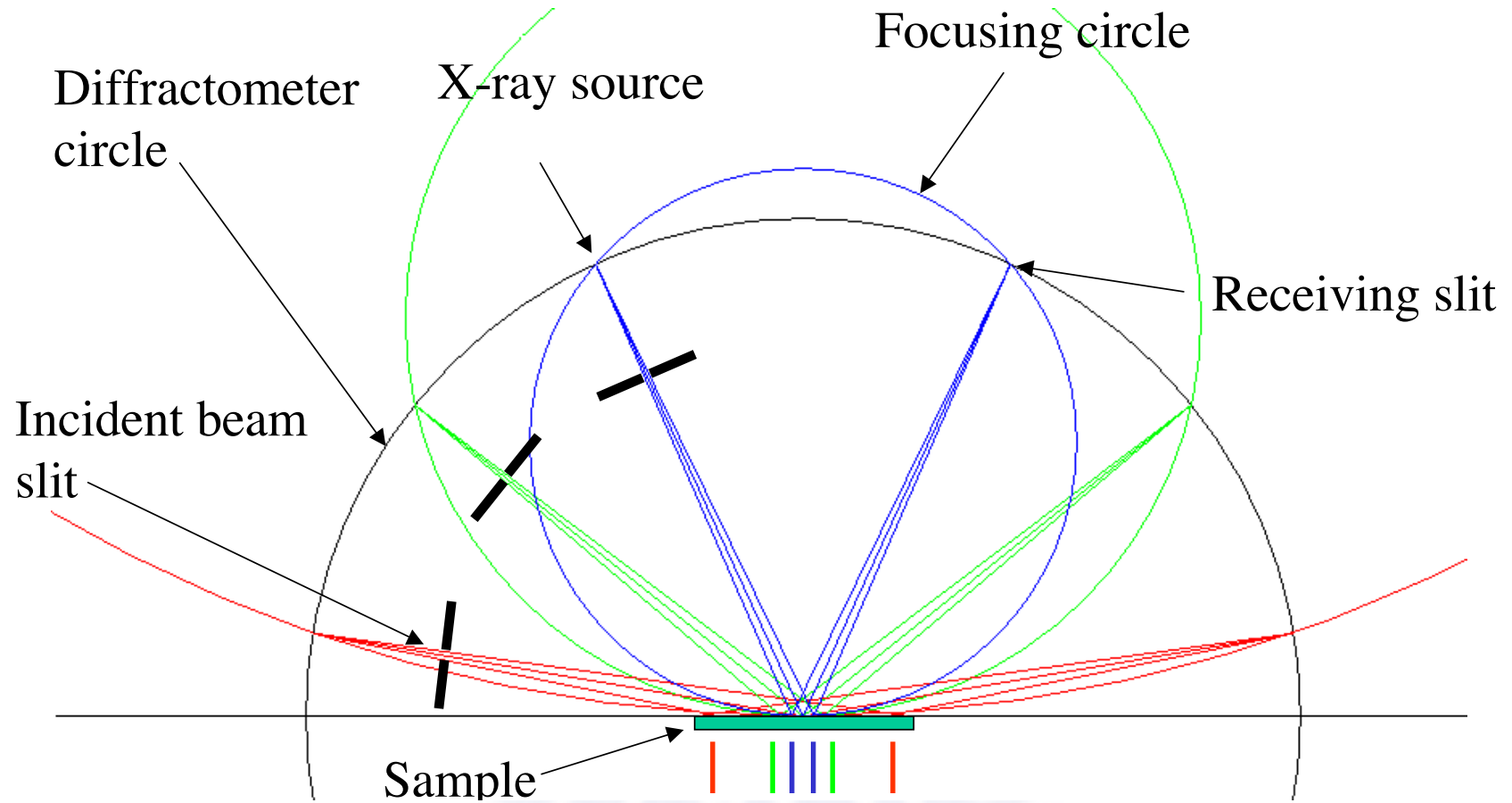
Bragg Brentano 2-THETA:THETA Setup



A typical spectrum

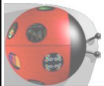
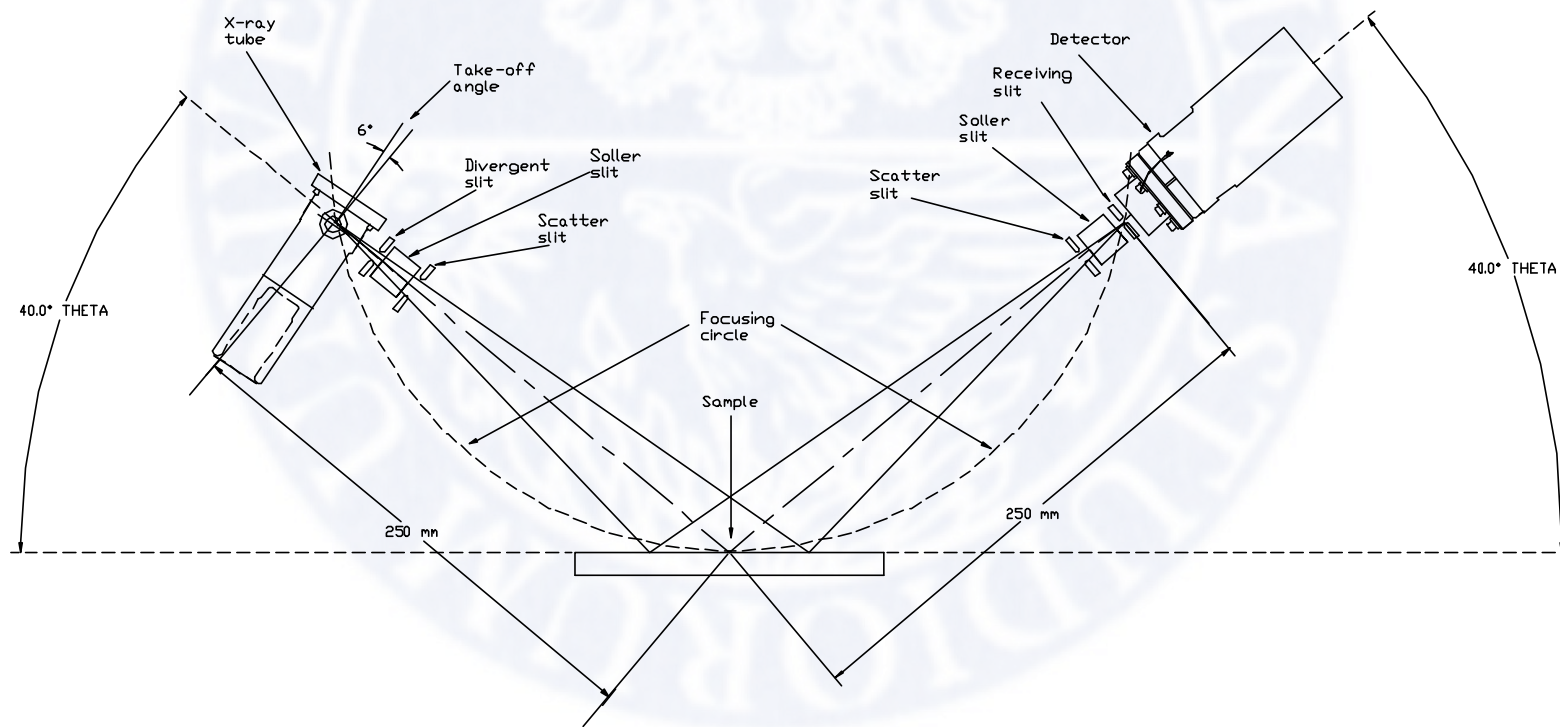


Parafocusing circle (Bragg-Brentano)

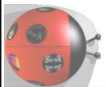
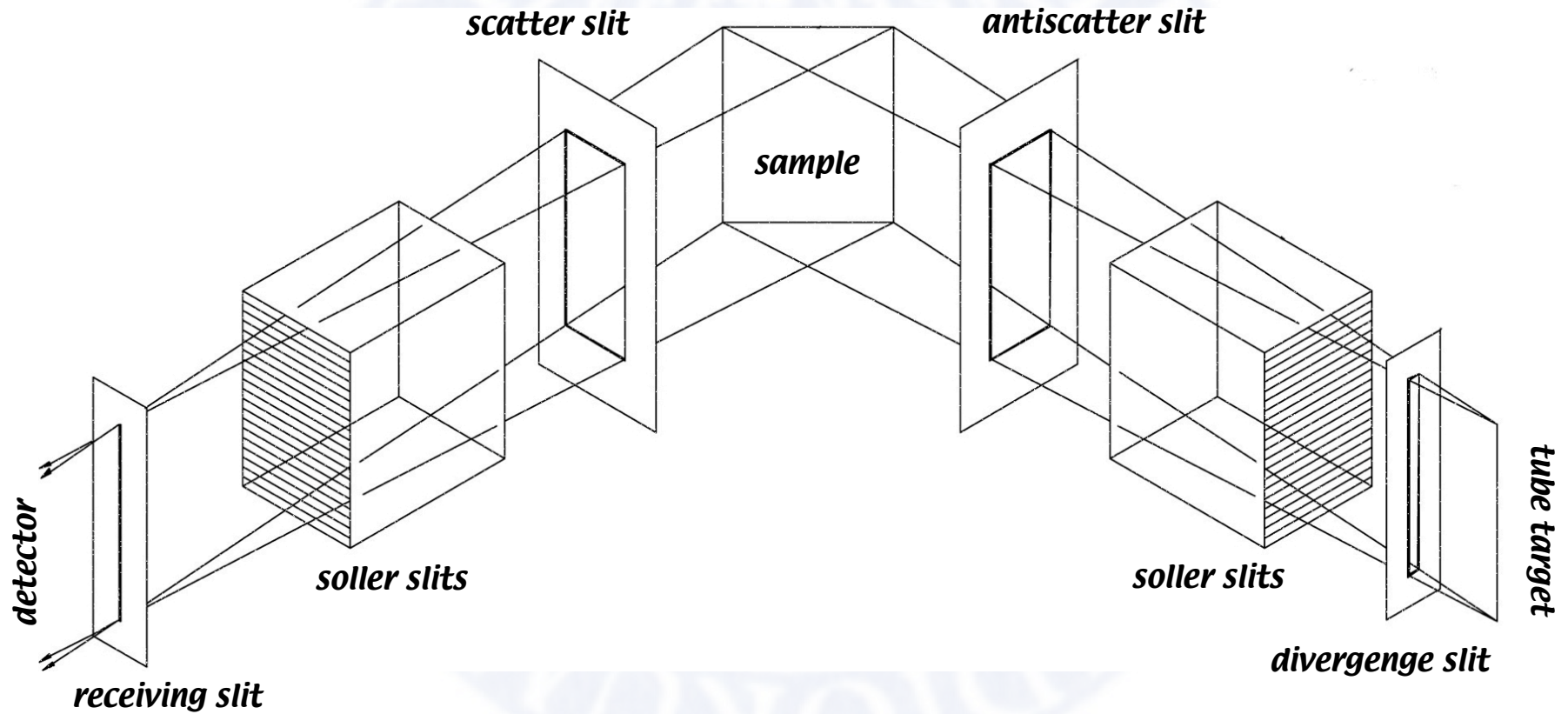


Theta-theta diffractometer

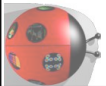
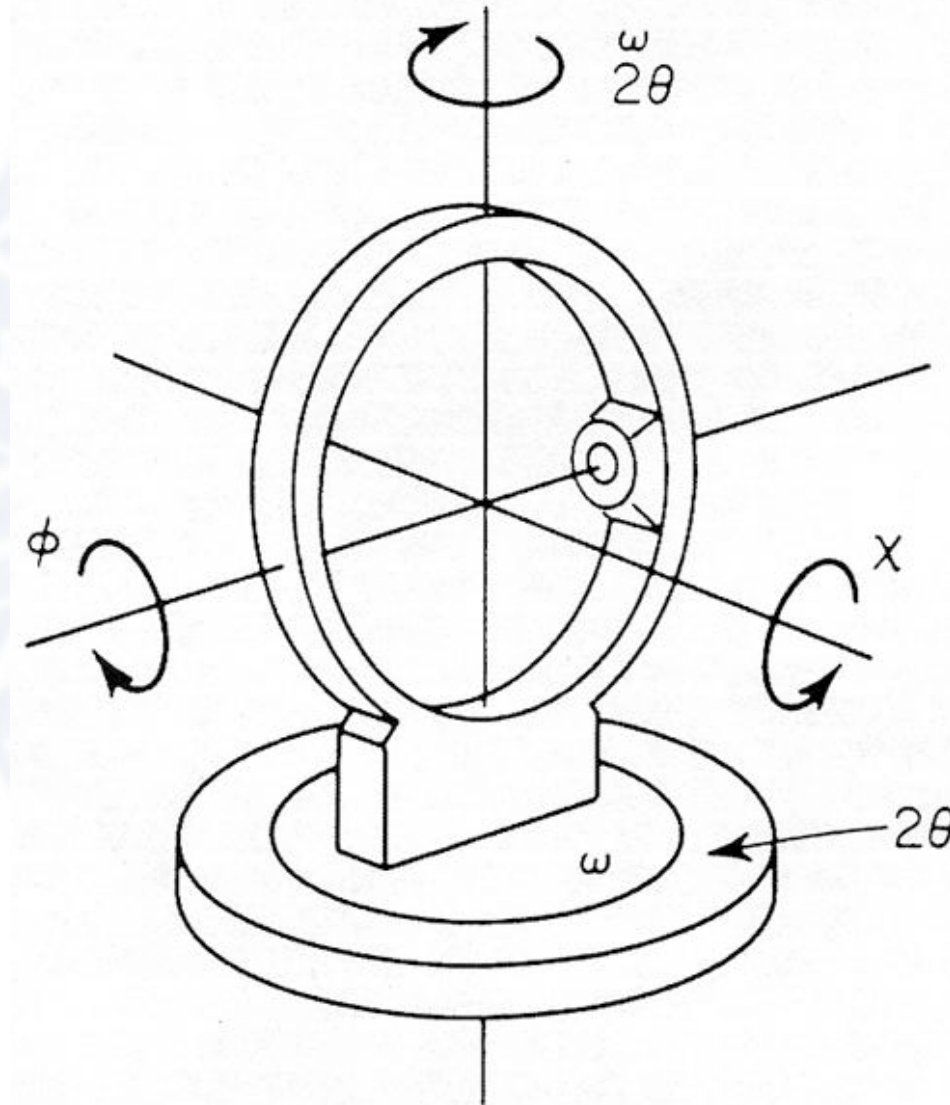
Bragg Brentano THETA:THETA Setup



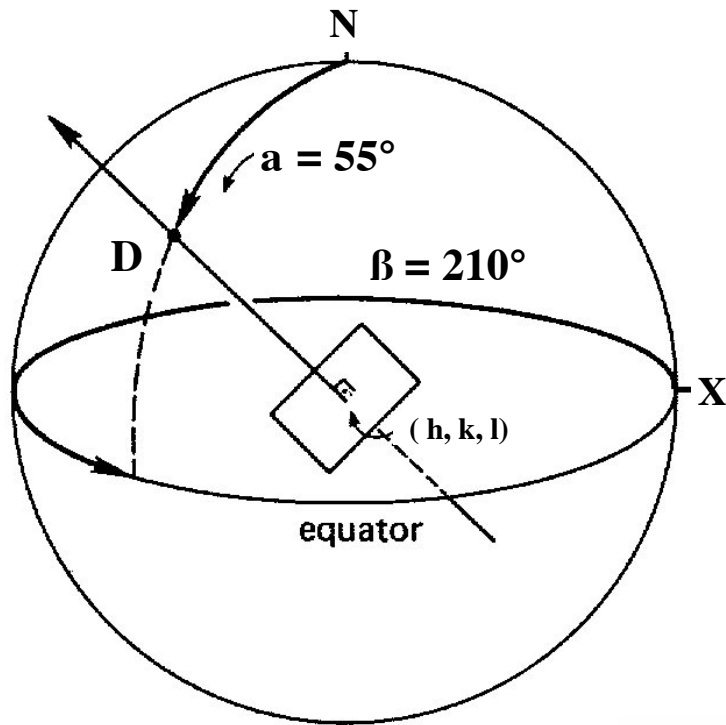
Slits system in Bragg-Brentano



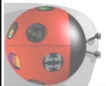
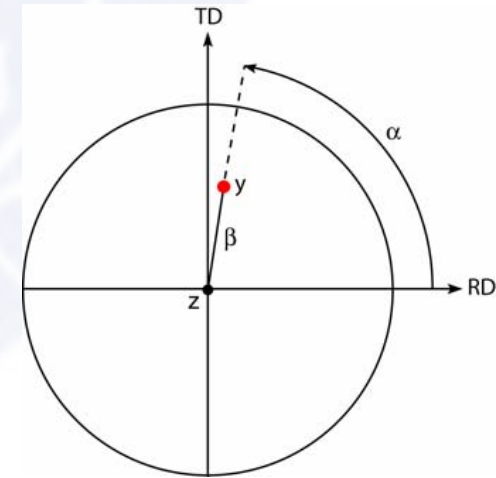
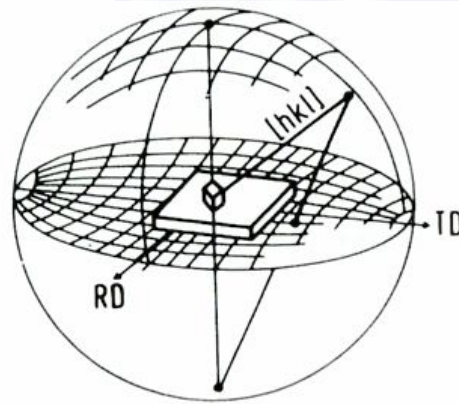
Texture goniometer



Texture orientations

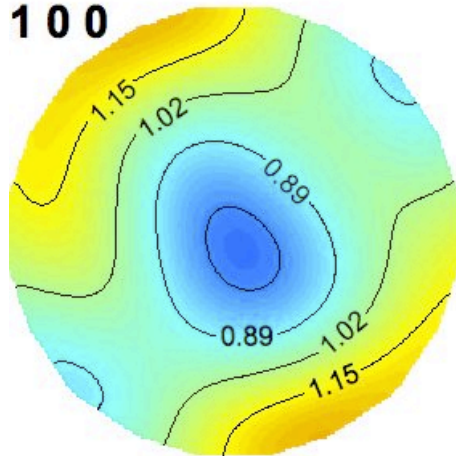


Pole figure representation

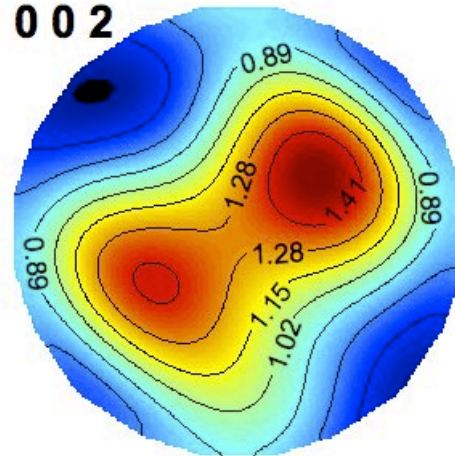


Pole figure projections (and inverse)

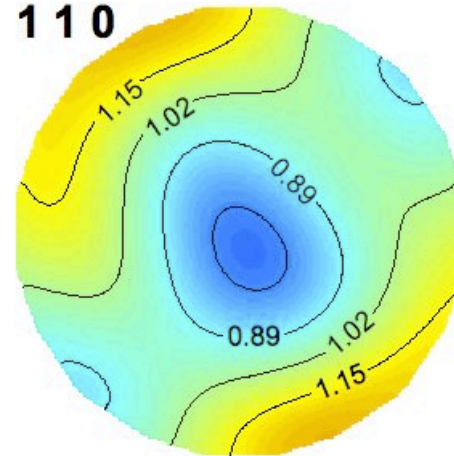
1 0 0



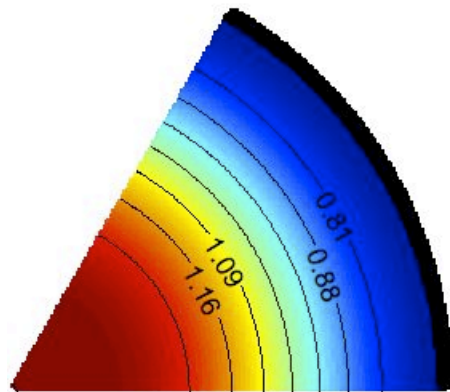
0 0 2



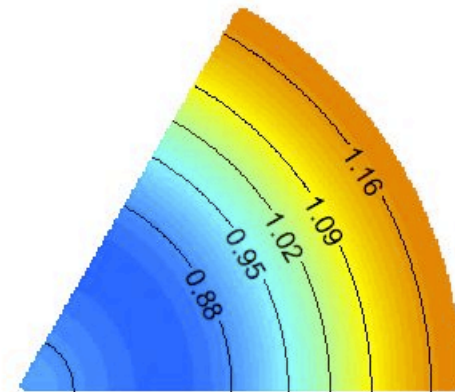
1 1 0



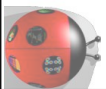
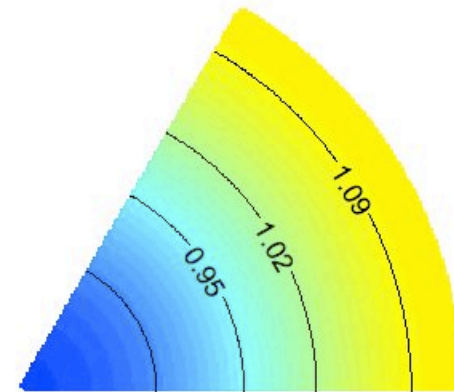
ND



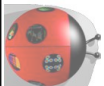
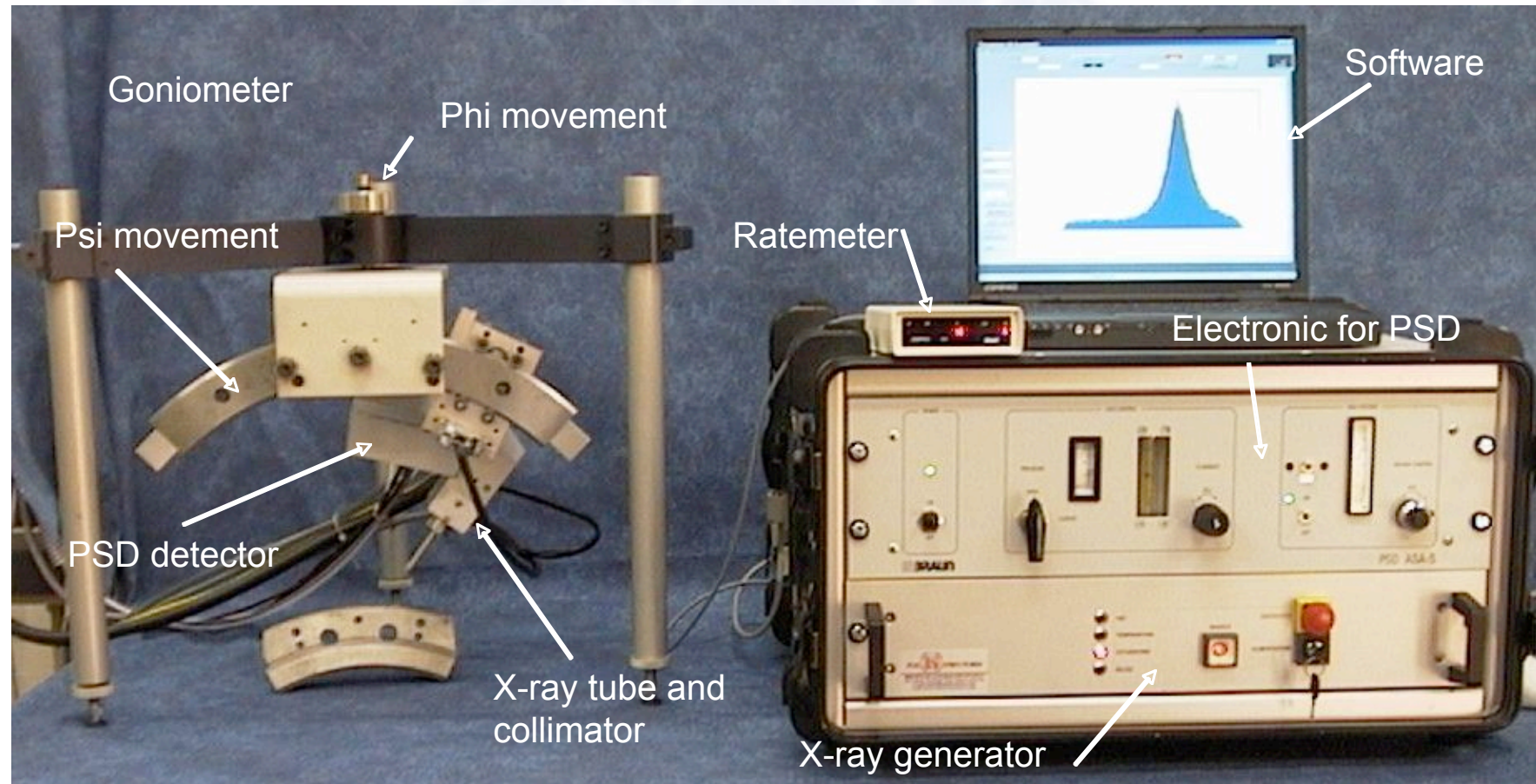
RD



TD

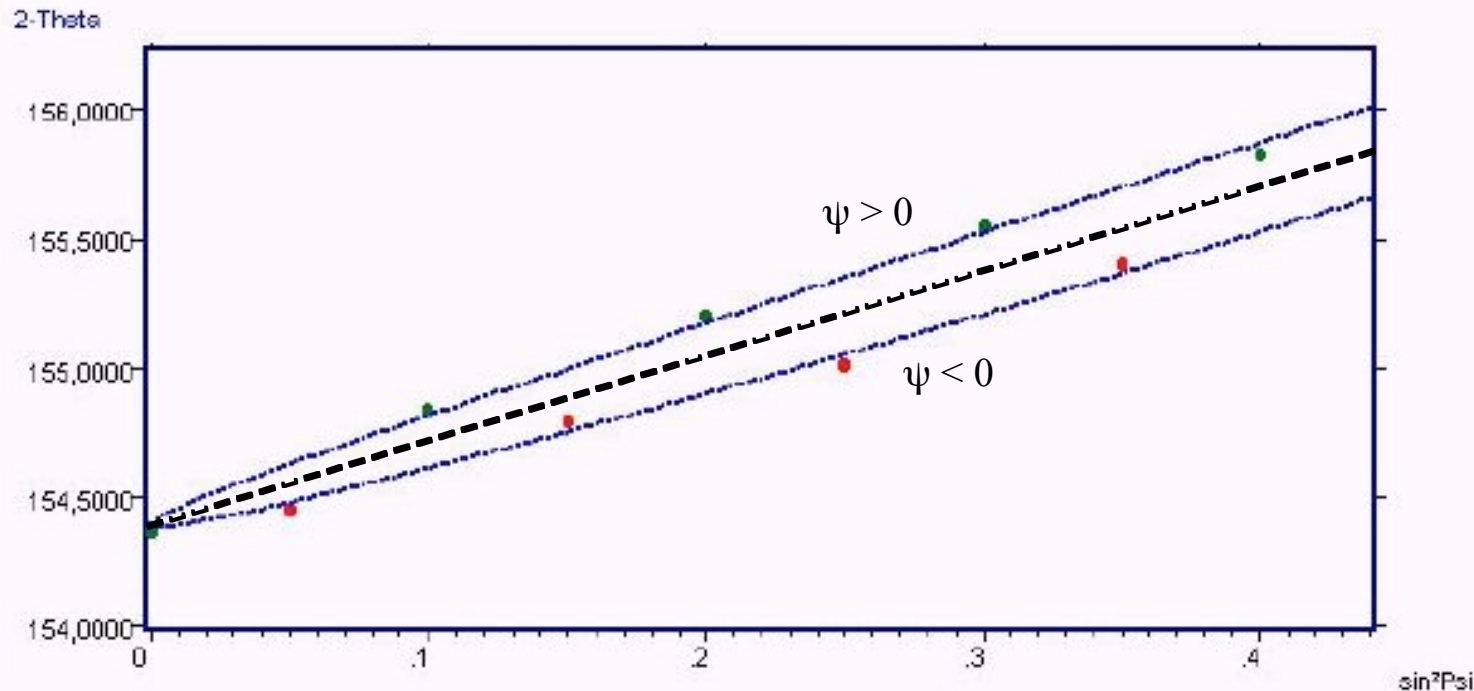


Residual stress measurement

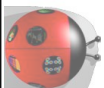


Residual stress analysis

campione M04/045/2 zona A-1 residual stress -1082 MPa +/-60 MPa

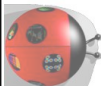


When either or both of ϵ_1 , ϵ_2 are non-zero, d measured at positive and negative Ψ will be different due to the argument $\sin^2\Psi$ associated with these terms causing a "split" in the d (2-theta) vs. $\sin^2\Psi$ data. This effect is termed Ψ -splitting.



Diffraction analyses

- *Phase identifications (crystalline and amorphous)*
- *Crystal structure determination*
- *Crystal structure refinements (cell parameters and atomic positions)*
- *Quantitative phase analysis (and crystallinity determination)*
- *Microstructural analyses (crystallite sizes - microstrain distributions etc.)*
- *Texture analysis*
- *Residual stress analysis*
- *Order-disorder transitions and compositional analyses*
- *Thin films*



Search-Match and the PDF system

